



Institut Charles Gerhardt Montpellier

CHEMISTRY: MOLECULES TO MATERIALS



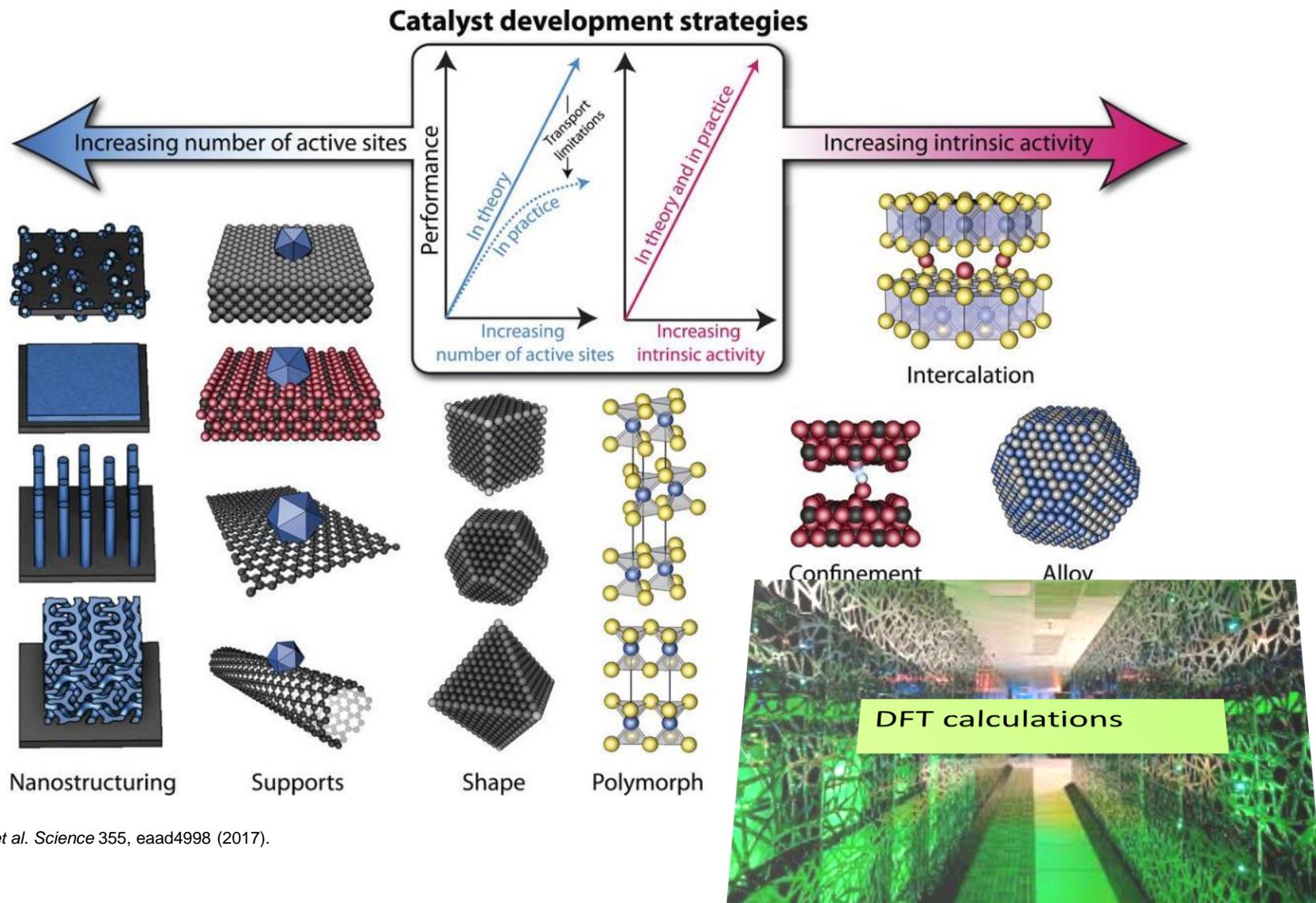
Toward an Accurate Theoretical Description of Gold-based Nanocatalysts Under Reactive Media

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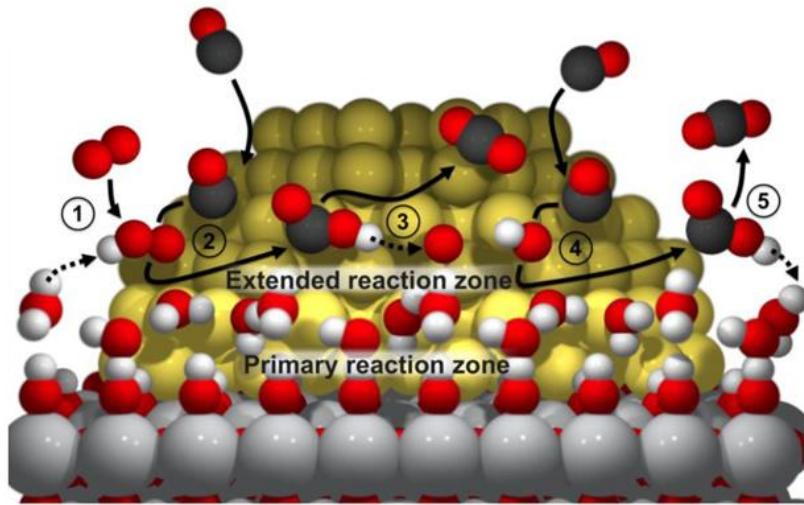
The context: Theoretical description of nanocatalysts



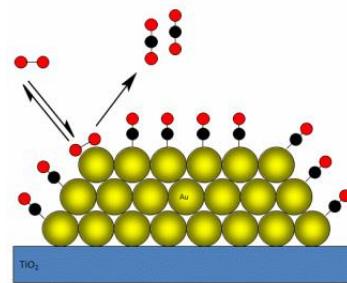
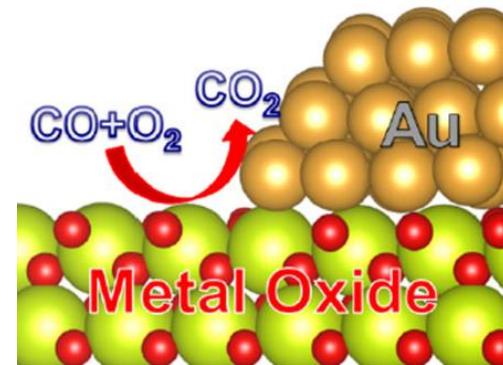
Seh, Z. W. et al. Science 355, eaad4998 (2017).

Subject: Gold based nanocatalysts under reactive media

Problematic: Gold shows remarkable catalytic activity for CO oxidation even at room temperature. Despite extensive studies, the mechanism of catalysis by gold nanoparticles is still unclear, in particular in relation to CO oxidation at room temperature.

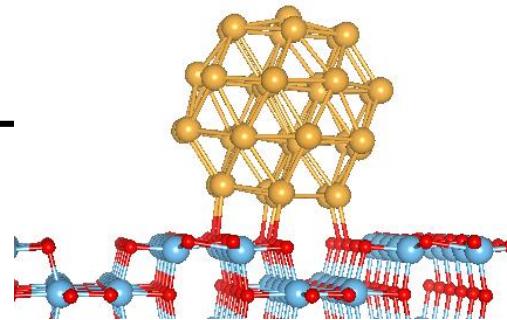


Chandler B.D. et al. Science, 2014, 345, p. 1599.

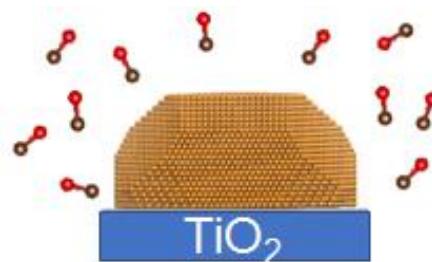


Goal: Structure and morphology changes under working conditions

- Modeling supported Au-Cu nanoparticles



- Identification of the shape evolution of GNP under O₂ gas: MSR model



Modeling supported Au-Cu Nanoparticles

Context: Optimization of Au-Cu catalysts

Goal: to model “realistic” systems: supported catalysts

Main questions:

✓ The effect of the support

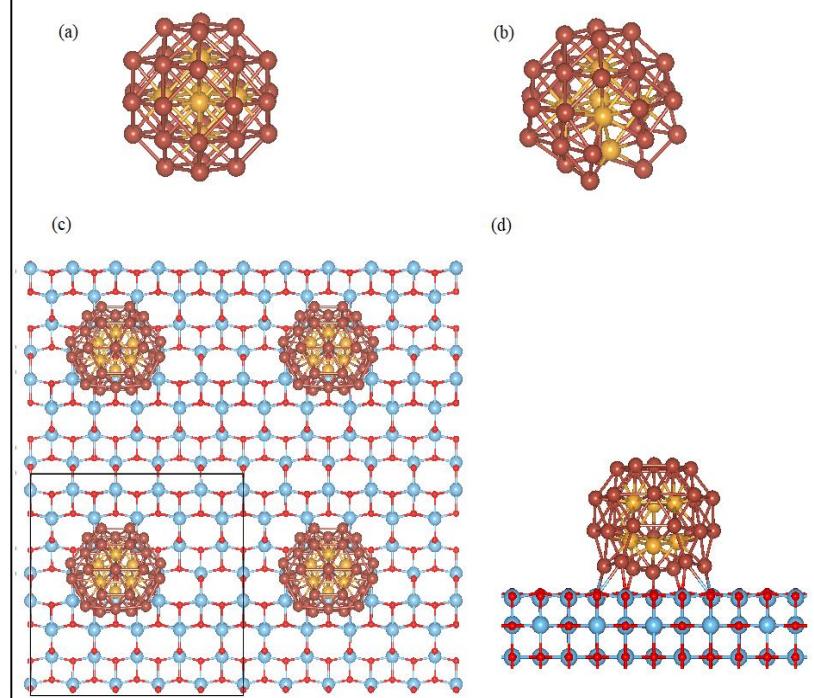
(small nb of atoms/Nbands ; large nb of Kpoints)

✓ The effect of the gas

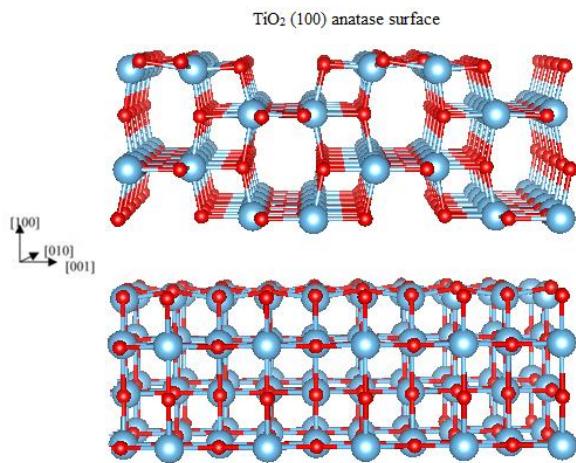
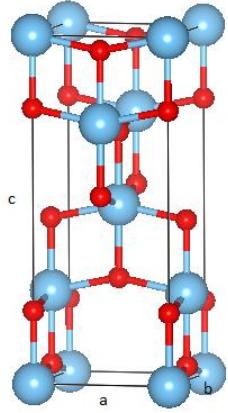
(Large nb of atoms/Nbands ; small nb of Kpoints)

✓ The minimum energy structures of AuCu nanoalloys : Hybrid functionals

(non easy convergence)



Modeling Au-Cu : The effect of the support



From the bulk to the surface

The Hubbard correction is used for the description of strongly correlated materials:

In titanium oxide, the 3d orbitals do not overlap with the oxygens on the titanium atoms. Thus an appropriate semi-empirical approach based on the Hubbard model leads to an effective Hamiltonian to treat the

repulsion. In the Hubbard model, the electron-electron repulsion parameter U . The DFT+ U total energy is given as the sum of the correction term that depends on the orbital occupation numbers.

$$E^{DFT+U}[\rho, n_s] = E^{DFT}[\rho] + E^U[n_s]$$

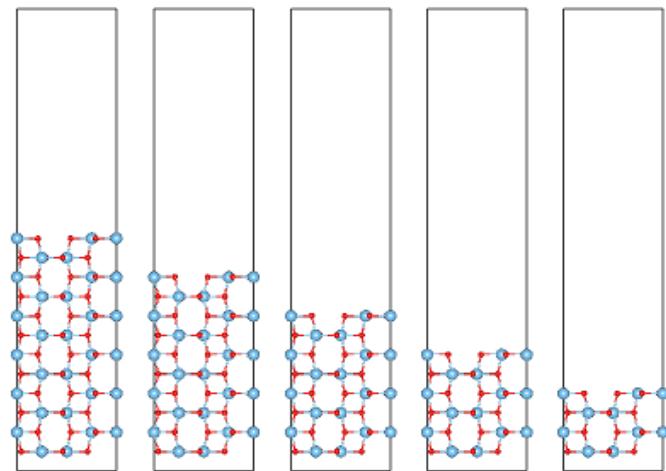
In the U total electronic energy, the contributions of core electrons are counted. This is the reason why one has to include the following:

$$E^U[n_s] = E_0^U[n_s] - E^{dc}[n_s]$$

(2.7)

where $E_0^U[n_s]$ is a term derived according to the HF treatment of electron-electron interaction of electrons within the shell s and $E^{dc}[n_s]$ is a double counting term which represents the electron-electron interaction contribution of the shell s that is already present.

Surface energy calculations following the Fiorentini and Methfessel method: $f(x)=ax+b$
 $b= (E_{\text{slab}} - E_{\text{bulk}})/2x A$



2x1 surface slab = KPOINT (33x13x1)

Methodology

Optimisation of TiO₂(100) surface

+

Correction de Hubbard

119 irreducible k-points

Nodes=6 :NPAR=4/KPAR=6

4 layers: 48 (16 Ti + 32 O) atoms : NBANDS= 232

5 layers: 60 (20 Ti + 40 O) atoms : NBANDS= 288

6 layers: 72 (24 Ti + 48 O) atoms : NBANDS= 352

7 layers: 84 (28 Ti + 56 O) atoms : NBANDS= 408

8 layers: 96 (32 Ti + 64 O) atoms : NBANDS= 472

```
executed on      Linux_Intel date 2017.08.24 14:54:14
running on 168 total cores
distrk: each k-point on 28 cores, 6 groups
distr: one band on NCORES_PER_BAND= 7 cores, 4 groups
```

Votre job (3433111) a déclenché un débordement mémoire. Il a tourné sur des nœuds qui disposent de 64 Go de mémoire. La machine Occigen dispose de nœuds avec 128 Go que vous pouvez adresser en rajoutant une option à la soumission de votre job (<https://www.cines.fr/calcul/faq-calcul-intensif/#tabs-2>). Attention, cette solution est réservée aux travaux justifiant ce besoin mémoire.



The problem was solved by considering Nnodes=KPOINTS

```
executed on      Linux_Intel date 2017.09.11 12:47:15
running on 2856 total cores
distrk: each k-point on 24 cores, 119 groups
distr: one band on NCORES_PER_BAND= 3 cores, 8 groups
```

Cost: 27 100 core.h !!!

Modeling Au-Cu : The effect of the reactive gas

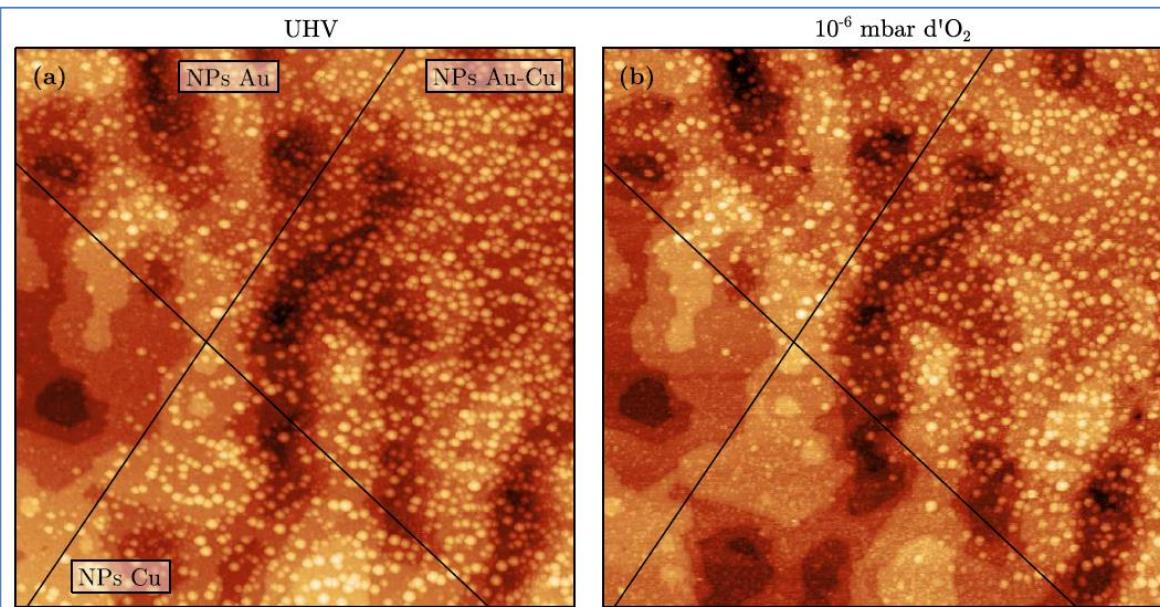
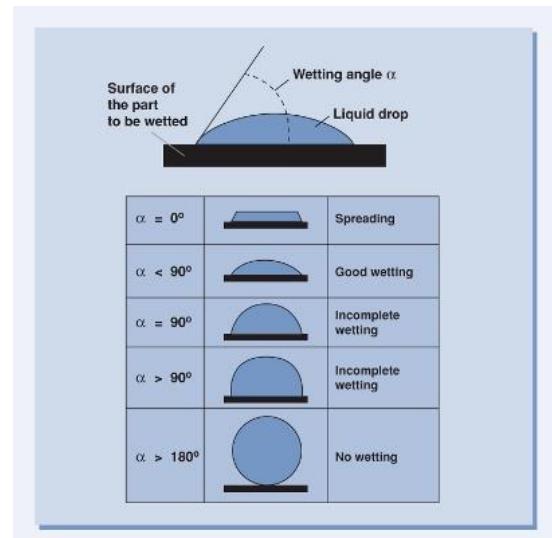


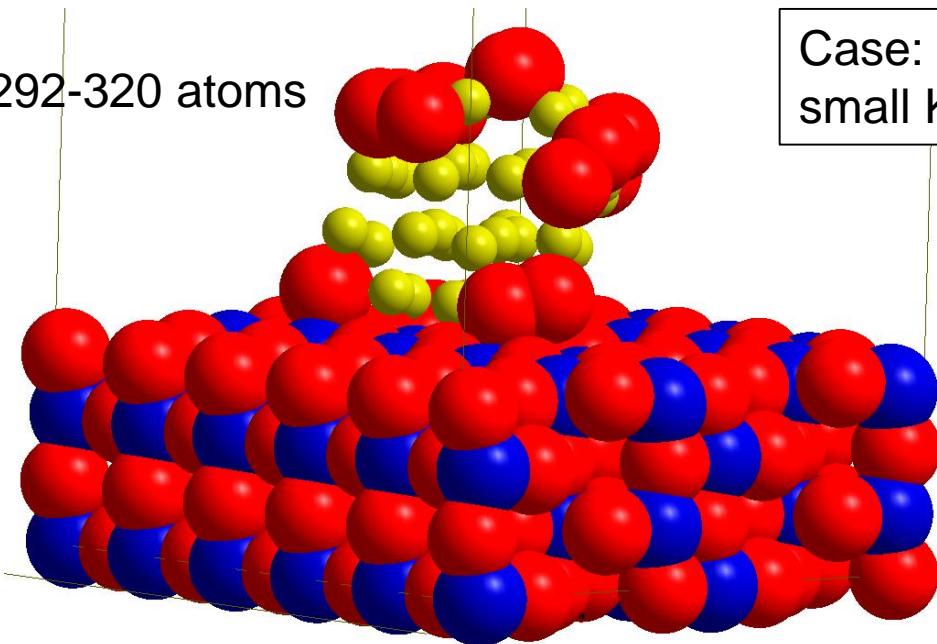
Figure 16 – Image STM $300 \times 300 \text{ nm}^2$ obtenues après une séquence d'évaporation Au puis Cu (a) et réalisée avec la méthode d'ombrage qui permet de synthétiser sur la même surface des NPs d'Au, de Cu et d'Au-Cu. Des changements se produisent pour les NPs de Cu après introduction de 10^{-6} mbar d'O₂ (b).



The supported Cu nanoparticles disappear under O₂ pressure: Wetting phenomena

Modeling Au-Cu : The effect of the reactive gas

292-320 atoms



Case: Large NBANDS/
small KPOINTS

POSCAR
1.000000000000000
19.105000000000004 0.000000000000000 0.000000000000000
0.000000000000000 19.059000000000011 0.000000000000000
0.000000000000000 0.000000000000000 29.194500000000014
Ti O Cu O
80 160 38 14
Selective dynamics
Direct
0.000000000000000 0.0512922599999968 0.000000000000000 F F F
0.0999996500000009 0.3012928400000021 0.0654401800000031 F F F
0.0999996500000009 0.176292549999994 0.000000000000000 F F F
0.000000000000000 0.426293129999977 0.0654401800000031 F F F
0.000000000000000 0.551293309999984 0.000000000000000 F F F
0.0999996500000009 0.801293889999966 0.0654401800000031 F F F
0.0999996500000009 0.6762936000000010 0.000000000000000 F F F
0.000000000000000 0.926294179999993 0.0654401800000031 F F F
0.1999994800000024 0.051292259999968 0.000000000000000 F F F
0.2999991300000033 0.3012928400000021 0.0654401800000031 F F F
0.2999991300000033 0.176292549999994 0.000000000000000 F F F

Dimension of arrays:

k-points	NKPTS =	5	k-points in BZ	NKDIM =	5	number of bands	NBANDS=	1458
number of dos	NEDOS =	301	number of ions	NIONS =	292			
non local maximal	LDIM =	6	non local SUM 21+1	LMDIM =	18			
total plane-waves	NPLWV =	*****						
max r-space proj	IRMAX =	1731	max aug-charges	IRDMAX=	5046			
dimension x,y,z	NGX =	96	NGY =	96	NGZ = 144			

Computational cost

Pour nos systèmes comprenant plus de 300 atomes, les tests de calculs (load balancing et communication bandwidth) ont montré un débordement de mémoire pour les calculs avec moins de 480 cœurs. Une fois nous avons déterminé le nombre de nœuds qu'il nous fallait pour des questions de mémoires, nous avons effectué des tests pour l'efficacité (malheureusement pas beaucoup de tests de peur de gaspiller nos heures) et voici un exemple des résultats :

Nanoparticule de Au-Cu couverte de O₂ et déposée sur un support TiO₂ : 314 atomes, 1494 NBANDS

Sur 20 nœuds de 24 cœurs (480 cœurs) : un LOOP+ coute : $480*3891/3600$ soit 518 cores.hours ...
LOOP+: cpu time 3897.27: real time 3898.03

Sur 25 noeuds de 24 coeurs (600 coeurs) : un LOOP+ coute : $600*3562/3600$ soit 593 cores.hours
LOOP+: cpu time 3562.86: real time 3562.20

Sur 35 noeuds de 24 coeurs (840 coeurs): un LOOP+ coute : $840*996s/3600$ soit 232 cores.hours
LOOP+: cpu time 996.67: real time 1000.01

Un calcule qui converge c'est 147 LOOP+, ce qui fait 34 104 cores.hours

Cela veut dire qu'un calcul sur 840 cœurs coute deux fois moins de ressources que le même calcul sur 480 cœurs.

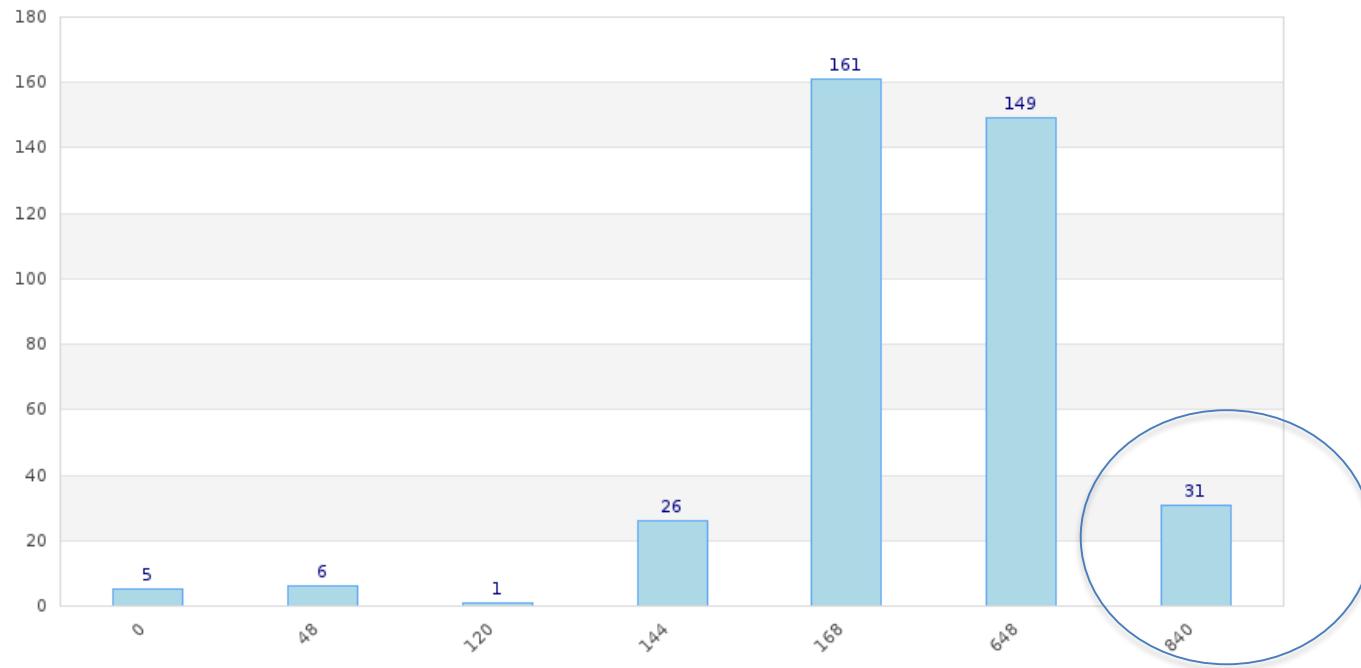
Pour plus de documentation sur le sujet voir le témoignage de Peter Larsson sur ce sujet.

<https://www.nsc.liu.se/~pla/blog/2014/01/30/vasp9k/>





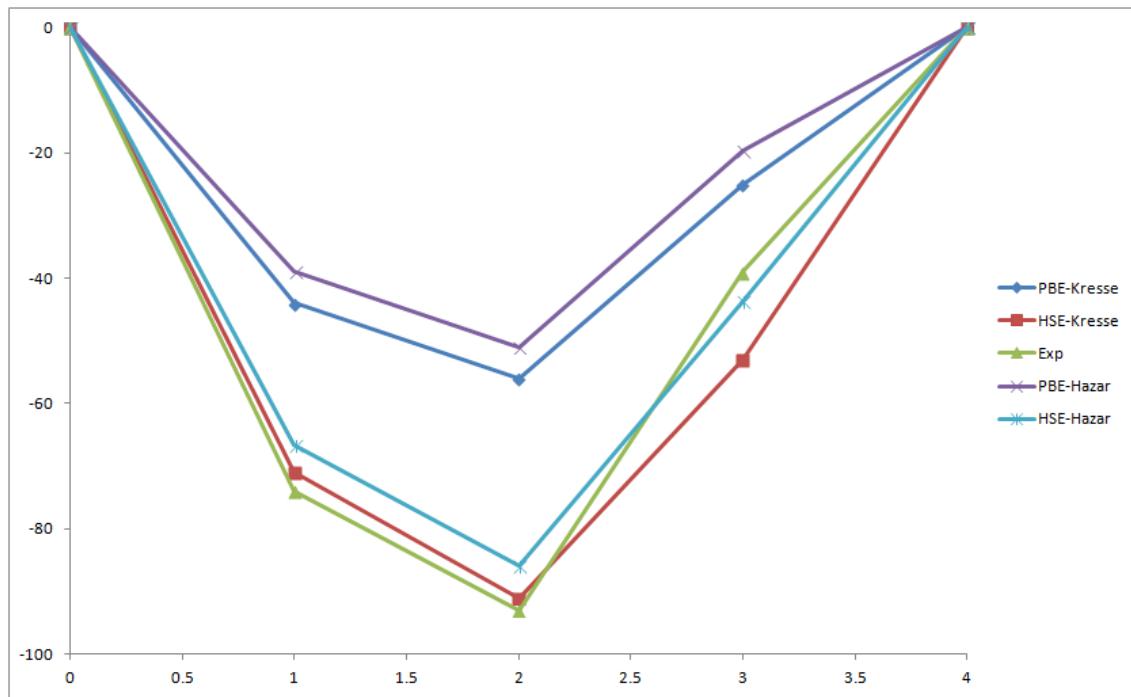
group lrs2684 du 2018-03-27 au 2018-08-21: nombre de travaux/coeurs - Nombre total de travaux : 379



$31 \times 10\ 000h = 310\ 000h$ (en 4 mois et demi)

Minimum energy structure of Au-Cu NPs: HSE06 vs. PBE

Calculation of the formation energies for the bulk

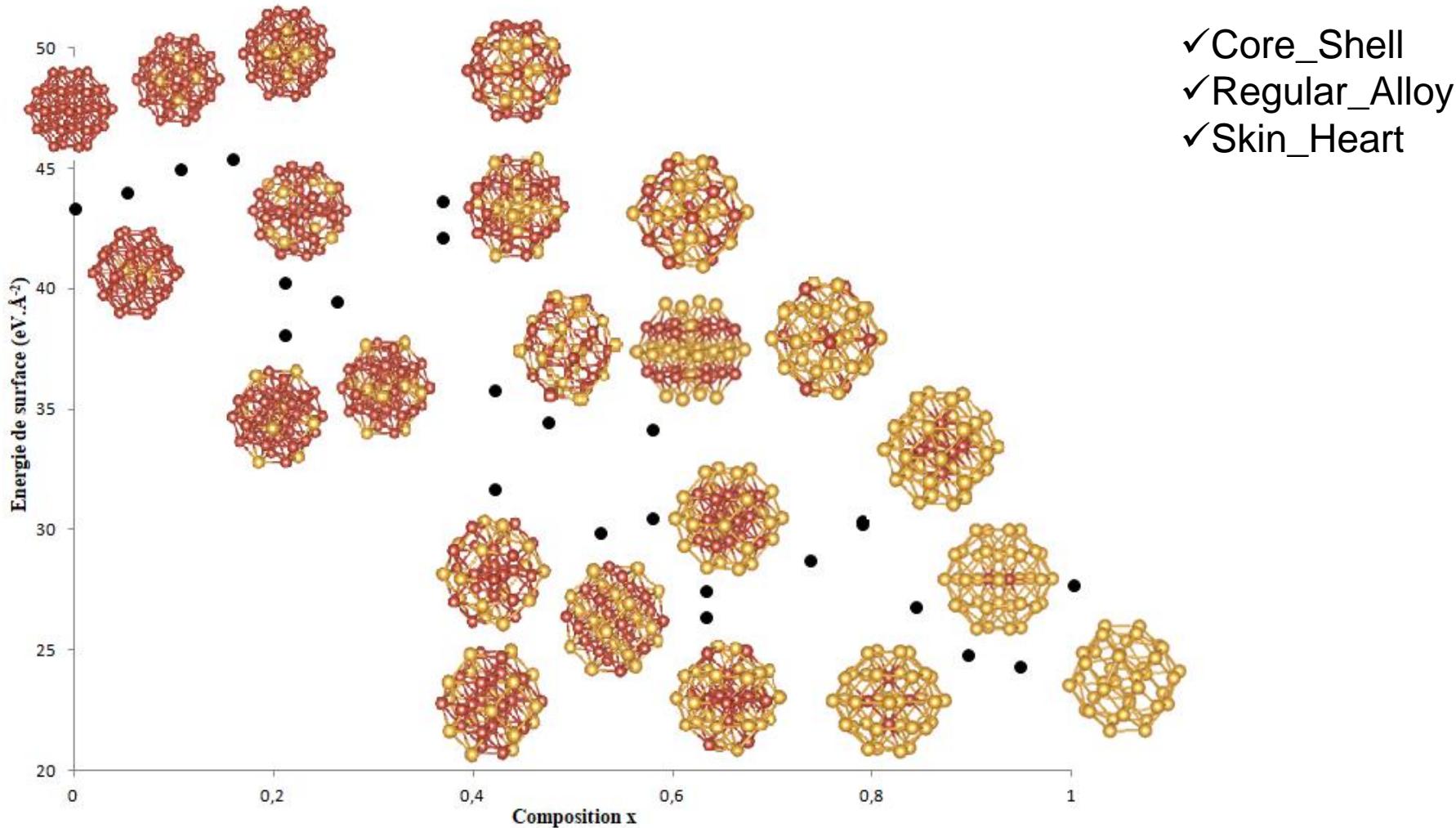


```
# -----
# Electronic minimization
# -----
PREC = NORMAL
# ALGO = Damped pour HSE06
ALGO = Damped
# ALGO = N
PRECFOCK = Fast
TIME = 0.4
# EDIFF = 1e-6
EDIFF = 1e-8
ENCUT = 600
LHFALC = .TRUE.
HFSCREEN = 0.2
# -----
# Ionic relaxation
# -----
NSW = 301
```

HSE06 gives results in very good concordance with experimental results

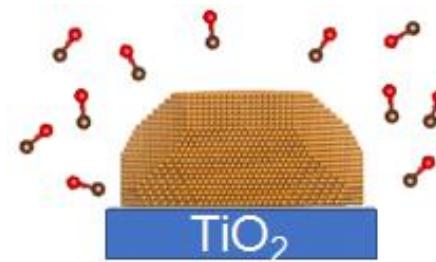


Stability of different AuCu alloy types:





- Identification of the shape evolution of GNP under O₂ gas: MSR model



State of the art : Structure and morphology changes under working conditions

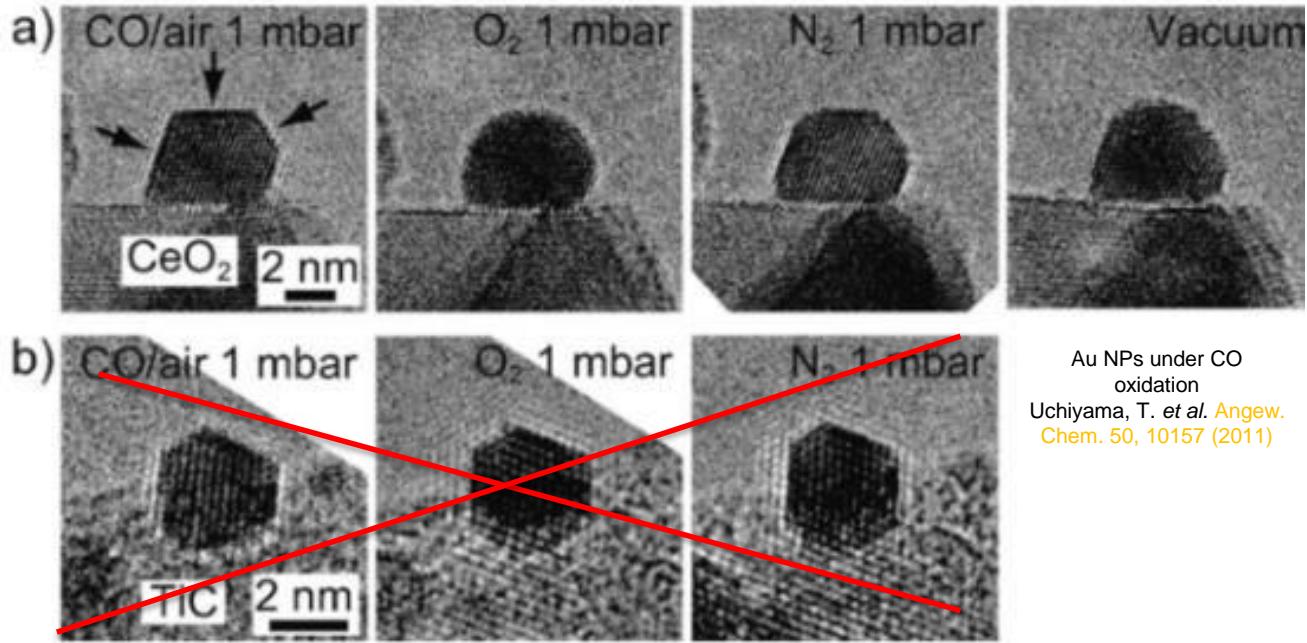
GNP supported on CeO₂ in various environments at room temperature.

⇒ During CO oxidation in CO/air gas mixture, the GNP appeared to be faceted in the form of a stable polyhedron enclosed by the major {111} and {100} facets.

⇒ Unexpectedly, the GNP behaved differently, and became rounded in pure O₂ gas. The GNP exhibited major facets in both inactive N₂ gas at 1 mbar and in vacuum.

⇒ GNPs supported on crystalline TiC remained polyhedral in all of the gas environments including pure O₂ gas.

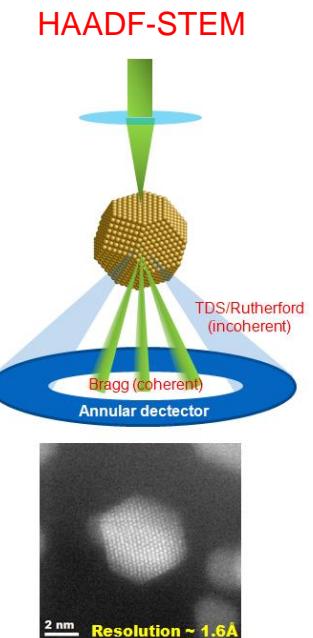
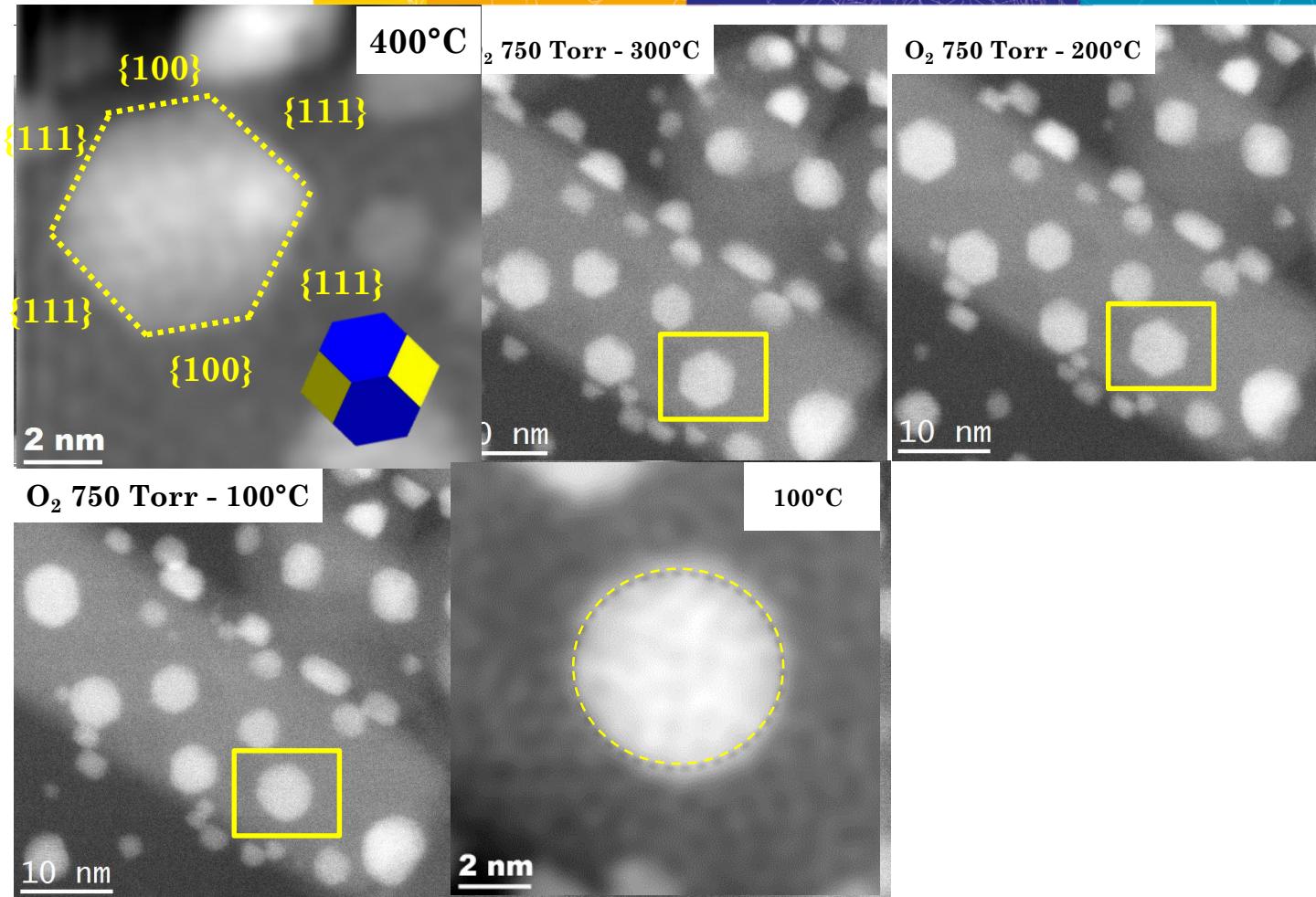
Gold nanoparticles



Au NPs under CO oxidation
Uchiyama, T. et al. *Angew. Chem.* 50, 10157 (2011)

What could be the link between reactivity and morphology changes ?

State of the art : Structure and morphology changes under working conditions



M P Q
~~~~~

Jaysen Nelayah and Adrian Chmielewski

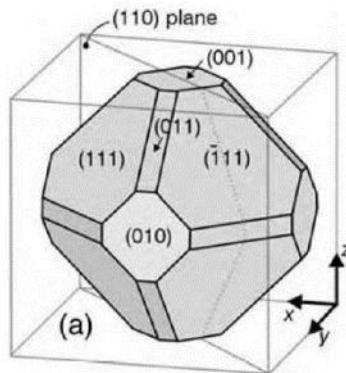
$373 \text{ K} < T < 273 \text{ K}$  : Rounding of the NPs

# Identification of the shape evolution of GNP under O<sub>2</sub> gas

## Equilibrium crystal shape: the Wulff construction

Under vacuum

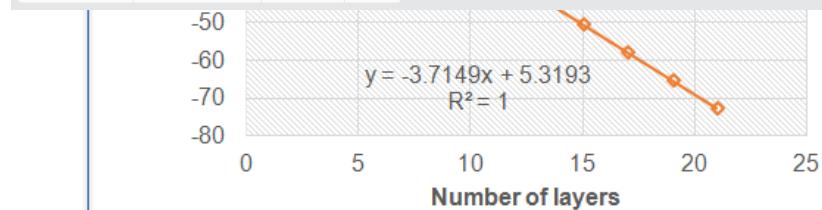
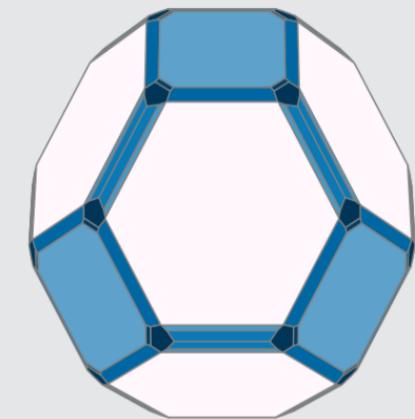
- The distance of a surface plane from the center-of-mass of the crystal is proportional to the surface energy of this plan



$$d_{hkl} = C \gamma_{hkl} \quad (c \text{ is a constant})$$

DFT calculations of  $\gamma_{hkl}$

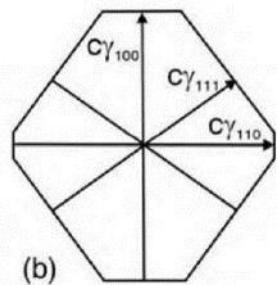
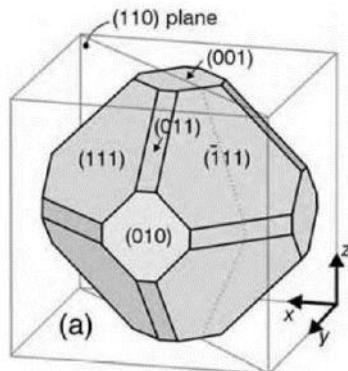
| Miller Indices (hkl) | Surface Energy (J/m <sup>2</sup> , eV/Å <sup>2</sup> ) | Area Fraction | Slab (CIF) |
|----------------------|--------------------------------------------------------|---------------|------------|
| (111)                | 1.31, 0.08                                             | 0.60          |            |
| (332)                | 1.43, 0.09                                             | 0.00          |            |
| (322)                | 1.45, 0.09                                             | 0.00          |            |
| (100)                | 1.47, 0.09                                             | 0.20          |            |
| (221)                | 1.48, 0.09                                             | 0.03          |            |
| (331)                | 1.52, 0.09                                             | 0.06          |            |
| (311)                | 1.54, 0.10                                             | 0.08          |            |
| (110)                | 1.56, 0.10                                             | 0.00          |            |
| (321)                | 1.58, 0.10                                             | 0.00          |            |
| (310)                | 1.59, 0.10                                             | 0.01          |            |
| (210)                | 1.60, 0.10                                             | 0.03          |            |
| (320)                | 1.62, 0.10                                             | 0.00          |            |
| (211)                | 1.63, 0.10                                             | 0.00          |            |



# Identification of the shape evolution of GNP under O<sub>2</sub> gas

## Equilibrium crystal shape: the Wulff construction

- The distance of a surface plane from the center-of-mass of the crystal is proportional to the surface energy of this plan



$$d_{hkl} = C \gamma_{hkl} \quad (c \text{ is a constant})$$

Under vacuum      Under gaz

$$d^{int}_{hkl} = C \gamma^{int}_{hkl}$$

$$\gamma_{hkl}^{int} = \gamma_{hkl} + \theta(E_{hkl}^{ads} / A_{hkl}^{at})$$

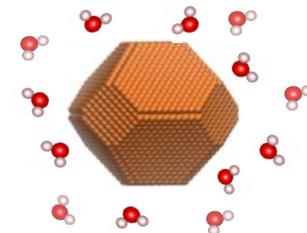
$\gamma_{hkl}$ : surface energy under Vacuum (DFT)

$E_{hkl}^{ads}$ : adsorption energy (DFT)

$A_{hkl}^{at}$ : surface area per atom (DFT)

$\theta$  is ( $T, P, E_{hkl}^{ads}$ ) dependent, described by Langmuir isotherm

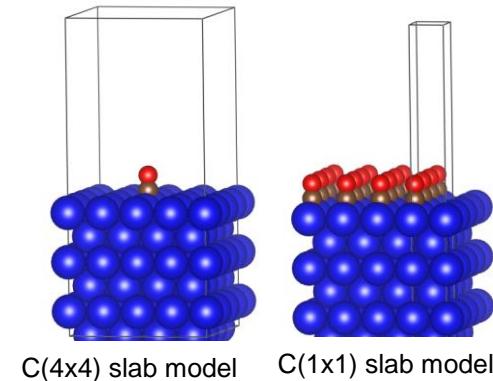
$$\frac{\theta}{1-\theta} = PK = P \exp\left(-\frac{\Delta G}{k_b T}\right) = P \exp\left(-\frac{E_{hkl}^{ads} - T(S_{ads} - S_{gas})}{RT}\right)$$



# Identification of the shape evolution of GNP under O<sub>2</sub> gas

Including the **lateral interaction** between adsorbates

$\theta$  is described by the **F-G isotherm**:



$$PK = \frac{\theta}{1 - \theta} e^{-c\theta}, c = \frac{zw}{RT}$$

$$K = \exp\left(-\frac{\Delta G}{k_b T}\right) = \exp\left(-\frac{E_{ads} - T(S_{ads} - S_{gas})}{RT}\right)$$

$z$ : coordination number of the surface site

$w$ : lateral interaction (DFT)

$$\gamma_{hkl}^{int} = \gamma_{hkl} + \theta((E_{hkl}^{ads} - zw\theta)/A_{hkl}^{at})$$

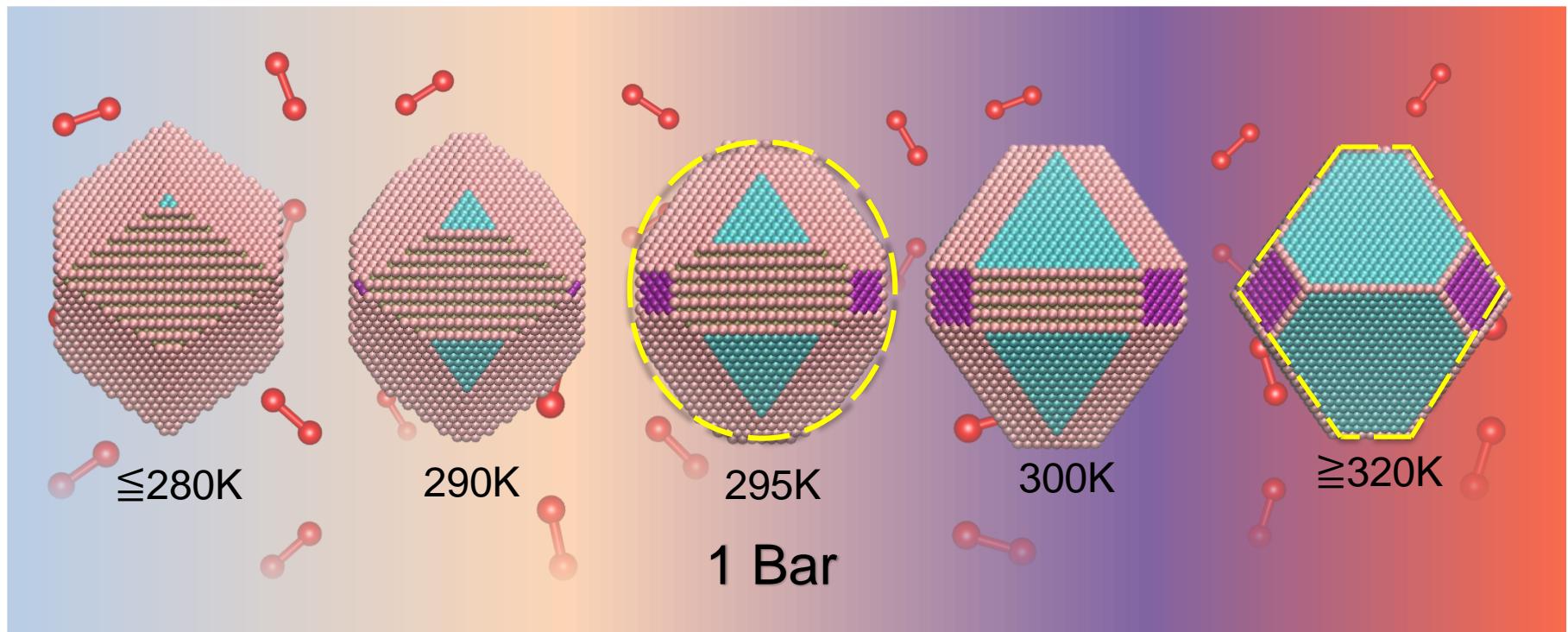
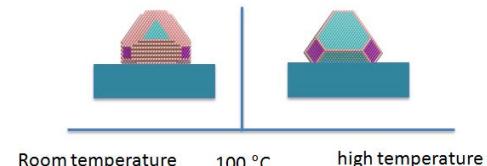
# Results: shape evolution of GNP under O<sub>2</sub> gas

DFT-D3 calculations of Eads O<sub>2</sub> molecule ( $d_{\text{Au-Au}}=4.07 \text{ \AA}$ ):

Au (111) = 0.00 eV

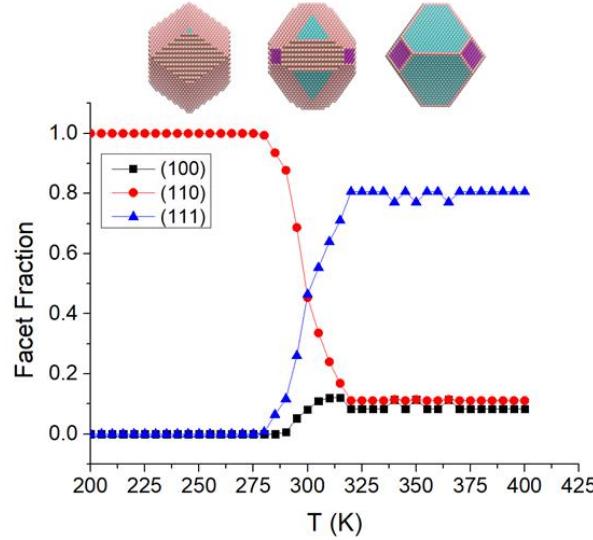
Au(100) = -0.22 eV

Au(110) = -0.34 eV

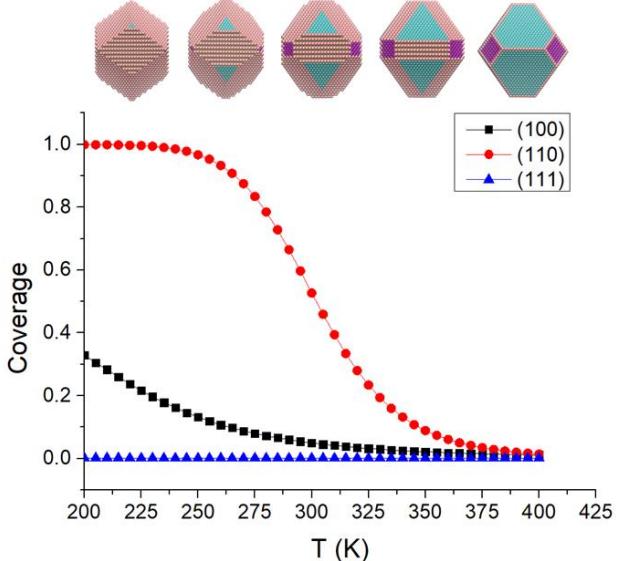


# Results: shape evolution of GNP under O<sub>2</sub> gas

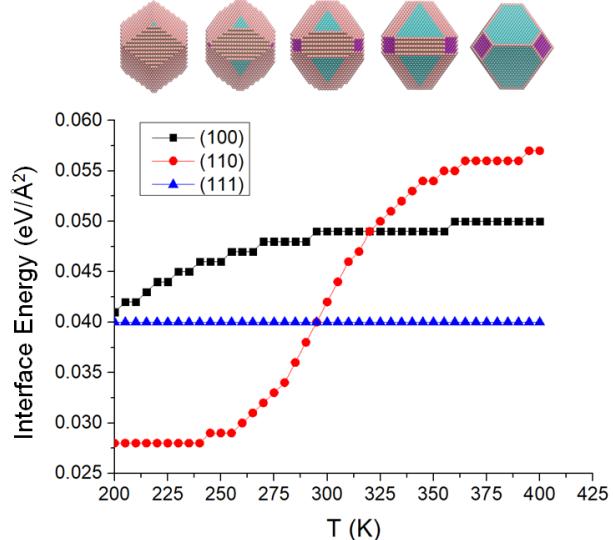
Facet fractions



O<sub>2</sub> coverage

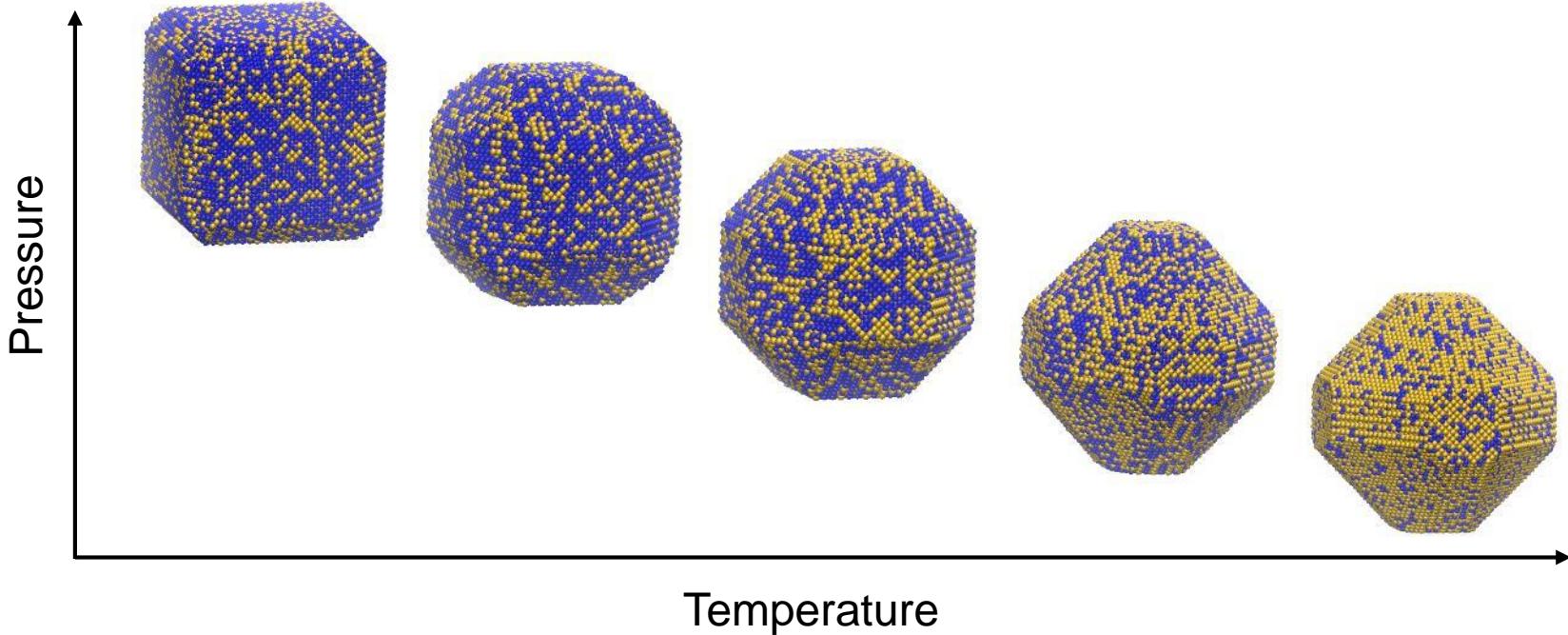


Interface energy



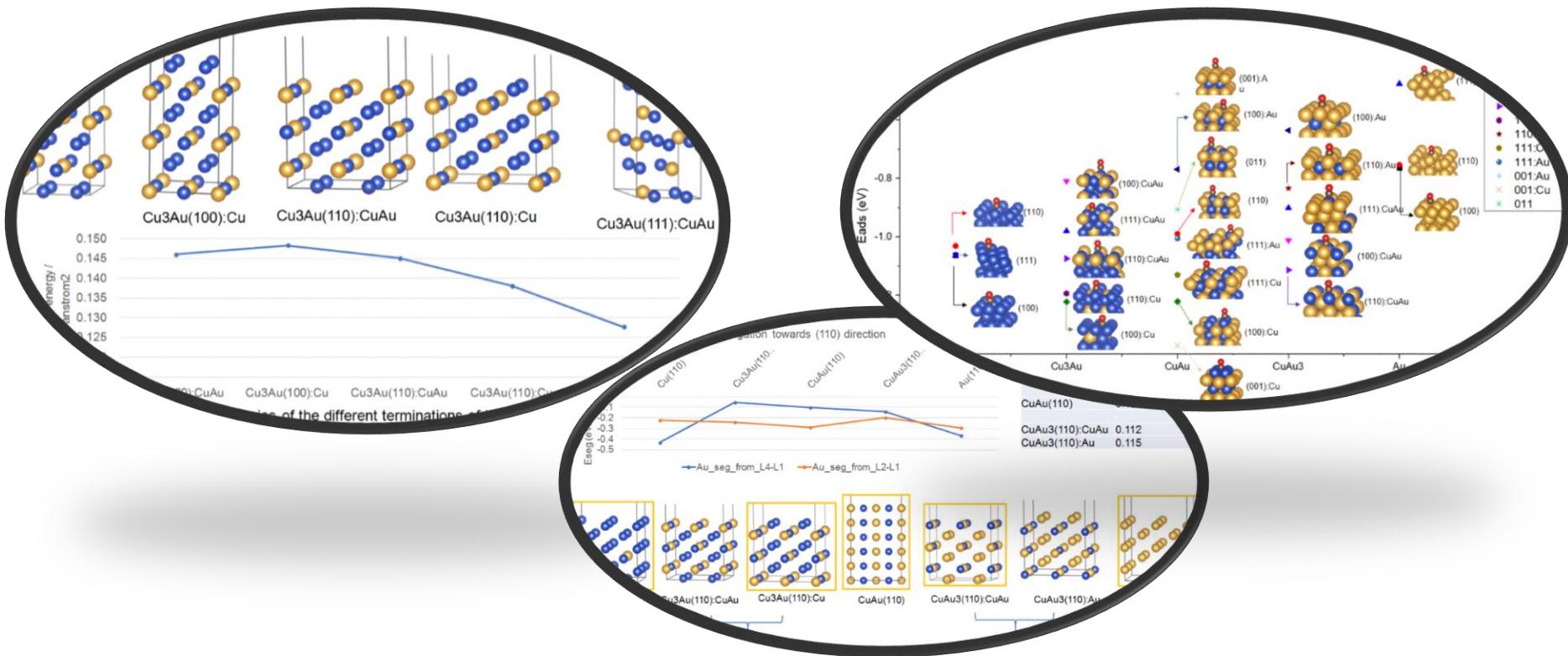
Evolution of active sites as a function of T

# Ongoing calculations: MSR model for nanoalloys



Ongoing work: Modeling Au-Cu NPs under CO gas

# Ongoing: MSR model for nanoalloys



$$\gamma_{alloy}^{int} = \gamma_{alloy} + f(\Delta H_{seg}) + f(\Delta H_{ads})$$

$\gamma_{alloy}^{int}$  is  $T$ ,  $P$ ,  $E_{seg}$ , and  $E_{ads}$  dependent!

# Methodology

```
SYSTEM =  
  
ISTART = 0  
ICHARG = 2  
PREC = Normal  
ISPIN = 2  
GGA = PE  
IVDW = 11  
  
ENCUT = 400  
EDIFF = 1E-5  
NELM = 200  
NELMIN = 8  
ISMEAR = 0  
SIGMA = 0.1
```

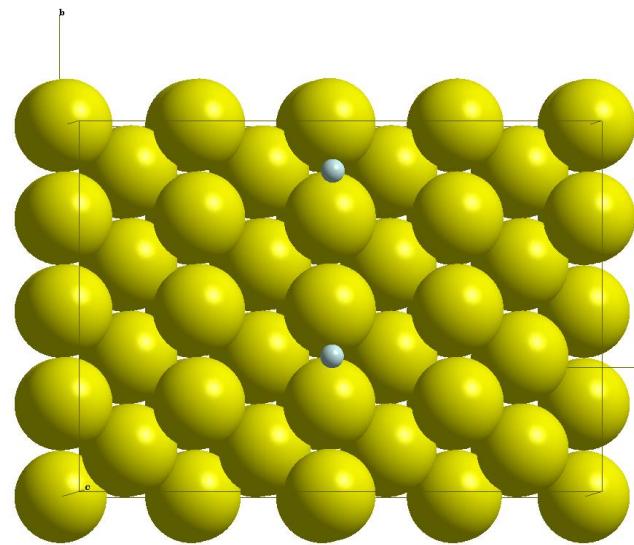
```
IBRION = 2  
ISIF = 2  
EDIFFG = -0.02  
NSW = 500  
ISYM = 0  
POTIM = 0.2
```

```
ALGO = F  
LDIAG = T  
LREAL = Auto
```

```
LWAVE = F  
LCHARG = F
```

```
NPAR = 2  
KPAR=2
```

```
#!/bin/sh  
#SBATCH --nodes=2  
#SBATCH --ntasks-per-node=28  
#SBATCH --ntasks=56  
#SBATCH --time=24:00:00  
#SBATCH --exclusive  
#SBATCH --output vasp.5.4.4.%J.out  
#SBATCH --constraint=BDW28  
##SBATCH --qos=bonus  
  
module load purge  
  
module load intel/18.0 openmpi/intel/2.0.1  
module list  
  
ulimit -s unlimited  
export OMPI_MCA_coll_tuned_use_dynamic_rules=1  
export OMPI_MCA_coll_tuned_alltoall_algorithm=4  
export OMPI_MCA_coll_tuned_bcast_algorithm=6  
  
srun --mpi=pmi2 -K1 --resv-ports  
/home/alloy/VASP.5.4.4/vasp.5.4.4/bin/vasp_ncl"
```



Au(110): 96 atoms Au +2H

```
vasp.5.3.3 18Dez12 (build Mar 13 2017 10:09:23) complex  
  
executed on          Linux_Intel date 2018.09.10 11:24:02  
running on      56 total cores  
distrk: each k-point on   28 cores,    2 groups  
distr: one band on NCORES_PER_BAND= 14 cores,    2 groups
```

## Dimension of arrays:

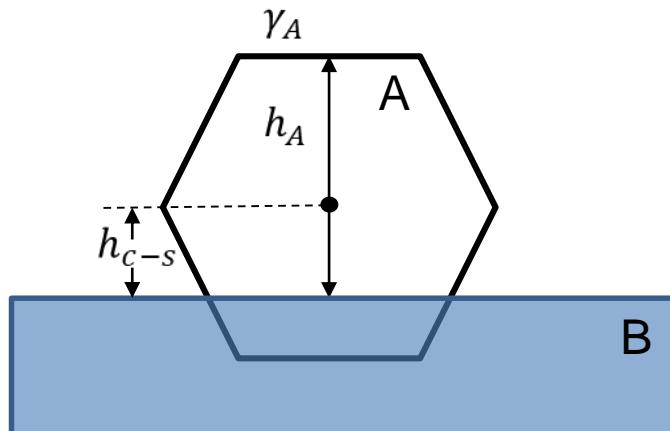
|                   |         |        |                    |         |    |                 |                |     |
|-------------------|---------|--------|--------------------|---------|----|-----------------|----------------|-----|
| k-points          | NKPTS = | 2      | k-points in BZ     | NKDIM = | 2  | number of bands | <b>NBANDS=</b> | 682 |
| number of dos     | NEDOS = | 301    | number of ions     | NIONS = | 96 |                 |                |     |
| non local maximal | LDIM =  | 6      | non local SUM 21+1 | LMDIM = | 18 |                 |                |     |
| total plane-waves | NPLWV = | 564480 |                    |         |    |                 |                |     |

# Computational cost

| Job_description                                                      | Number of atoms | Nb cores/job | Nb runs (x24h)      | Calculation time in cpu |
|----------------------------------------------------------------------|-----------------|--------------|---------------------|-------------------------|
| Optimizations of surfaces :<br>(111), (100), (110),<br>(221), (331)  | 54 et 108 atoms | 168/56       | 2-3 run per surface | 30 720                  |
| Adsorption of O <sub>2</sub> on (111), (100),<br>(110), (221), (331) | 64 et 118 atoms | 168/56       | 3-6 run per system  | 46 080                  |

# MSR model for Supported Metal NPs

- **Wulff-Kaischew theorem:** the distance from the center to the contact surface  $h_{AB}$  is proportional to the contact surface energy:  $\gamma_{c-s}$



$$\frac{\gamma_A}{\gamma_{c-s}} = \frac{h_A}{h_{c-s}} \quad \gamma_{c-s} = \gamma_{AB} - \gamma_B \\ \gamma_{AB} = \gamma_A + \gamma_B - E_{adh}$$

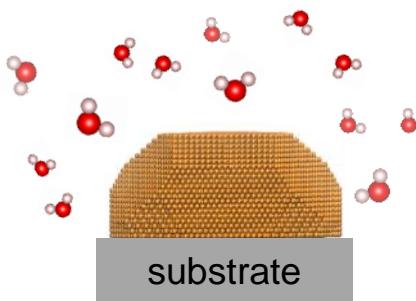
$\gamma_{AB}$ : interface energy between A and B.

$E_{adh}$ : adhesion energy

- **Wulff-Kaischew in the environment:**

$$\gamma_{A(hkl)}^{int} = \gamma_{A(hkl)} + \theta_A \frac{E_{A(hkl)}^{ads}}{A_{A(hkl)}^{at}}$$

$$\gamma_{c-s}^E = \gamma_{AB} - \gamma_{B(hkl)}^{int} = \gamma_{AB} - \gamma_B - \left( \frac{\theta_B E_B^{ads}}{A_B^{at}} \right) = \gamma_A - E_{adh} - \left( \frac{\theta_B E_B^{ads}}{A_B^{at}} \right)$$



$\theta_A, \theta_B$  are described by Langmuir isotherm

# DFT calculations of system with 700 atoms !! YES VASP CAN!

Adhesion energy calculations:

Size mismatch → (13x3) unit cell for Au(111)

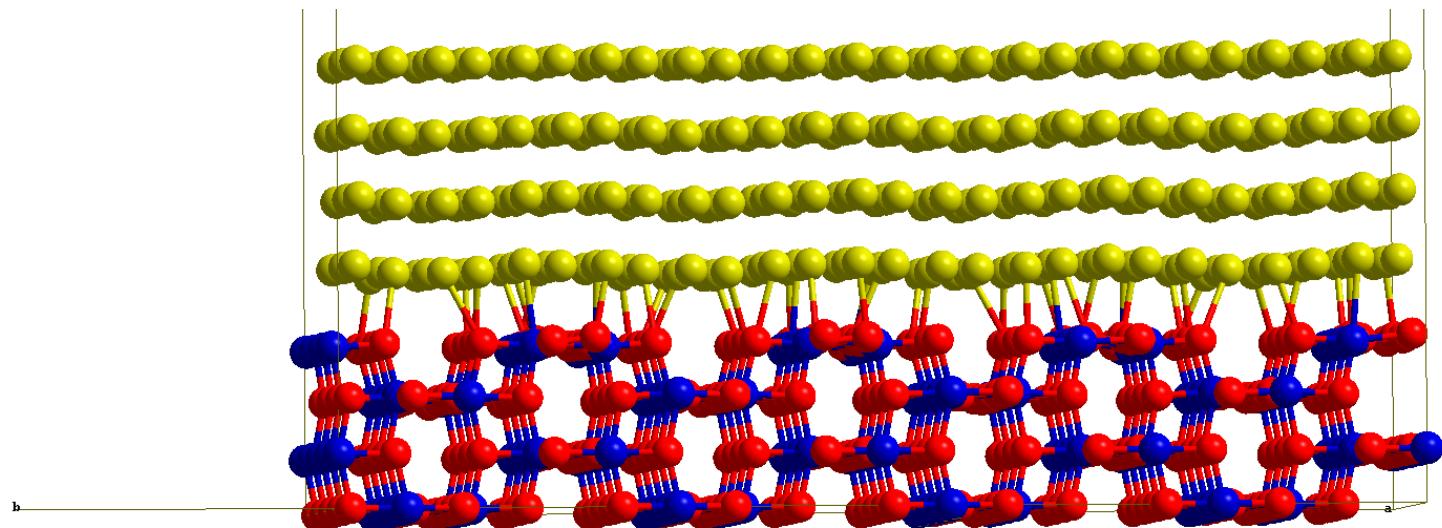
696 atoms

NBANDS=3920

K=gama

35 nodes/840 cores

30 LOOP+ = 4752 core.h

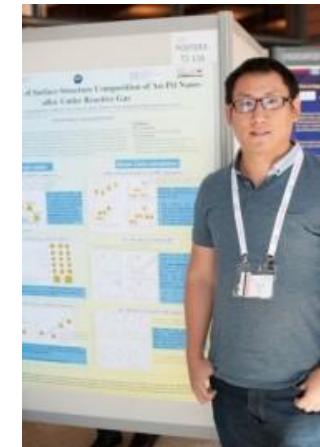




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Merci pour

votre attention

CHEMISTRY: MOLECULES TO MATERIALS

